

Langmuir–Blodgett Films of Heteropolyoxometalate/ Organomercury Acetylide Hybrid Composites: Characterization and Photoelectric Properties

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A new family of organometallic/inorganic nanohybrid Langmuir–Blodgett (LB) films consisting of rigid-rod organomercury acetylide complex (OMA-C₁₆ or OMA-C₈) as the π -conjugated organometallic molecule, tungsto(molybdo)phosphoric heteropolyacids HPA (HPA = H₃PMo₁₂O₄₀, H₃PW₁₂O₄₀, H₆P₂-Mo₁₈O₆₂, or H₆As₂W₁₈O₆₂) of the Keggin and Dawson structures as the inorganic component, and octadecylamine (ODA) as the auxiliary film-forming agent were prepared and characterized by π -A isotherms, UV–vis absorption spectra, fluorescence spectra, atomic force microscopy imaging, scanning tunneling microscopy, surface photovoltage spectroscopy, and low-angle X-ray diffraction. Our experimental results indicate that steady, even, and ordered Langmuir and LB films are formed in pure water and heteropolyacid subphase. They typically have a highly organized lamellar structure in which a monolayer of HPA is arranged alternately with well-ordered layers of OMA (and also ODA for OMA-C₁₆) and the periodicity of the layered structure for the OMA-C₁₆/ODA/H₃PMo₁₂O₄₀ LB film is ca. 3.7 nm. Luminescence spectra of hybrid LB films show that HPA can quench the emission of OMA-C₁₆ and OMA-C₈ to some extent. These LB composites show good photovoltage responses and a photovoltage of 103 μ V can be obtained for the OMA-C₁₆/ODA/H₃PMo₁₂O₄₀ system when it is excited by light at 330 nm. The monolayer LB films on silica wafer can also display interesting electrical conductivity behavior.

Introduction

The design, synthesis, and structural characterization of new hybrid materials, in which many applications can be predicted through the assembly of organic and inorganic building blocks, is an important current research area in materials science.¹ The construction of such organic/inorganic hybrids is regarded useful for obtaining multifunctional materials with new electrical and/or optical properties.² However, while most of the existing materials have been obtained as crystals, they find very limited practical applications. An alternative way in relation to the goal is to prepare

these materials as films or nanoparticles or to incorporate them into organic or inorganic matrices.³

Langmuir–Blodgett (LB) technique is considered an elegant approach to arrange molecules into well-organized multilayer films.⁴ It also allows the processing of molecules in the form of thin films, which are preferred for many applications and permits the control of the two-dimensional structure of these films at the molecular level along with ease of multilayer deposition, compared to other techniques such as spin-cast, vapor deposition, or self-assembled

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monolayers.^{4c,4d} For these reasons, the LB method has been widely applied to make ultrathin films with a specific architecture which can be exploited as chemo- or biosensors, modified electrodes, optical logic gates, or molecular electronic devices.^{4,5}

Polyoxometalate anions (POMs) represent a class of inorganic compounds that thanks to their topological and electronic versatilities have found wide applications in catalysis, biology, medicine, and materials science.^{6,7} Remarkably, they can act as electron reservoirs and are reversibly reduced readily by the addition of various numbers of electrons to give mixed-valence species. Their solubility and chemical stability in both aqueous and nonaqueous solvents make them very useful as the inorganic components of functional molecular materials.⁸

Since the work of Clemente-León and his co-workers in the preparation of inorganic/organic hybrid LB films using heteropolyacid and surfactant DODA (DODA = dimethyldioctadecylammonium) as film materials,⁹ research work in this field is continuously expanding and has mainly focused on the study of magnetic, electrochemical, and electrochromic properties of inorganic/organic hybrid LB films in which the organic surfactant is embedded in heteropolyacid with different shape, size, and charge.⁸ While the cationic surfactant plays only a structural role, the electronic properties of POM-based LB films are principally governed by the POMs. Another attractive method depends on creating novel hybrid LB films that combine the properties of both the inorganic entities and the organic component of the film. With this idea in mind, an organomercury acetylide complex (OMA) that can introduce electronic delocalization within the LB film was designed by us and it can be combined with electron reservoir POMs and/or amphiphilic octadecylamine (ODA) to prepare a new type of hybrid organometallic/inorganic LB films. While Langmuir and Langmuir–Blodgett films of a great variety of molecular metal complexes with interesting magnetic or optical properties have been prepared in the past few years,¹⁰ to our knowledge, the use of mercury-derived compounds in LB film prepara-

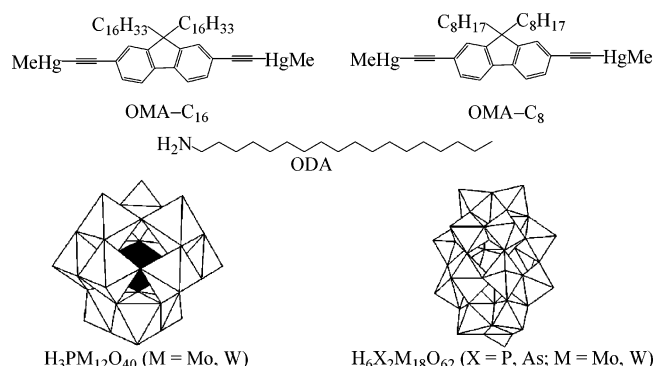


Figure 1. Molecular structure of compounds used in the construction of hybrid LB films.

tion is very rare.¹¹ As far as we know, self-assembly of group 12 mercury acetylide in forming LB film by the LB technique has not been reported previously. They are expected to have functional potential as a component of luminescent film devices. With our growing interests in the design and synthesis of luminescent functional materials of group 10–12 metallaynes,¹² we report here the first examples of the preparation of OMA/HPA hybrid LB films in the presence or absence of ODA and their structures and photovoltaic behavior are discussed in detail. A monolayer of ODA essentially serves as the auxiliary film-forming agent here and can be used as a template for adsorption of various POMs.

Experimental Section

Materials. Four types of HPA, viz. $H_3PM_{12}O_{40}$, $H_3PW_{12}O_{40}$, $H_6P_2Mo_{18}O_{62}$, and $H_6As_2W_{18}O_{62}$ (denoted as HPM_{12} , HPW_{12} , $HP2-Mo_{18}$, and HAs_2W_{18} , respectively) and the organomercury acetylide (OMA) coordinated with the 9,9-dioctylfluorene-2,7-diacetylide (OMA-C₈) and 9,9-dihexadecylfluorene-2,7-diacetylide (OMA-C₁₆) chromophores were synthesized according to the procedures previously reported.^{13,14} Octadecylamine and chloroform were used as received without further purification.

Monolayer and LB Film Fabrication. The formulas of the composites for the LB films are shown in Figure 1. HPA can be organized as monolayers using the LB technique. Monolayer formation and deposition were carried out on a French LB 105 slot under room-temperature condition at 20 ± 1 °C. The surface pressure was measured by the Wilhelmy method. Triple-distilled deionized water (pH = 6) was used as the subphase. The spreading solution of OMA-C₁₆ and ODA (1:1, 1×10^{-3} mmol dm⁻³) in

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chloroform was spread onto the pure water subphase using a microsyringe; when the solvent had evaporated thoroughly, compression began with the compression rate at $10 \text{ cm}^2 \text{ min}^{-1}$ and the curve was recorded. The procedure of LB film fabrication was essentially the same as that for the monolayer except that the subphase is HPA aqueous solution ($1 \times 10^{-6} \text{ mol L}^{-1}$). All the experiments for monolayer deposition were performed under a surface pressure of 30 mN m^{-1} . ODA plays only a structural role in ensuring a good-quality LB film for OMA-C₁₆ and it was observed that the organization of HPA molecules within the LB films is not so good without the ODA monolayer. But for OMA-C₈, we can obtain reasonable OMA-C₈/HPA LB films without the addition of ODA. In the case where a stable Langmuir monolayer of OMA-C₁₆/ODA or OMA-C₈ was formed on the subphase, the monolayer was subsequently deposited onto ITO or quartz substrates by the vertical dipping method at a rate of 1 mm min^{-1} , resulting in a fairly good deposition of a typical Y-mode film. The number of layers of LB film prepared here is equal to the number of dipping or lifting processes, on each of which a floating Langmuir monolayer was transferred onto the substrate with a good transfer ratio.

Instruments. Ultraviolet–visible (UV–vis) spectra were measured on a UNICAM Helios α spectrometer. Low-angle X-ray diffraction measurements were carried out on a Philips X'pert Pro instrument, operating with a monochromated Cu K α radiation source at 40 kV and 50 mA. Surface photovoltage spectroscopy was measured on a D-300 surface photovoltage instrument. Photoluminescence spectra were recorded on a SPEX F212 fluorescence spectrometer. AFM image and scanning tunneling microscopy of the LB films were measured on a SPA-400 atomic force microscope.

Results and Discussion

Synthesis. The organometallic mercury complexes OMA-C₁₆ and OMA-C₈ were prepared in good yields by the direct base-catalyzed mercuriation of 2,7-diethynyl-9,9-dihexadecylfluorene and 2,7-diethynyl-9,9-dioctylfluorene with 2 equiv of MeHgCl at room temperature according to the published procedures.^{14a,e} These compounds are readily soluble in chlorinated solvents such as CHCl₃ and CH₂Cl₂ to give a pale yellow solution. The long hydrocarbon chains are attached to the fluorene ring to enhance the solubility of the complex. The OMA complex shows a single ¹⁹⁹Hg NMR peak at around $\delta -453$ (versus $\delta -847$ for MeHgCl), reflecting polarization of the Hg–C \equiv bond.

Surface Pressure–Area Isotherms. Figure 2 shows the surface pressure–area (π – A) isotherms of OMA-C₁₆/ODA and OMA-C₈ on HPA and the pure water subphase solutions at 20 °C. The results show that they can form stable monolayer Langmuir film at the air–liquid interface. The molecular area of OMA-C₁₆/ODA and OMA-C₈ on the HPA aqueous solutions and pure water can be estimated by extrapolating the line part of π – A isotherm to the abscissa. The relevant data are displayed in Table 1. The isotherm is progressively modified by using different HPA polyoxoacids. For OMA-C₁₆/ODA film spread on the pure water subphase, for instance, the limiting area per molecule is 0.35 nm^2 , whereas that of OMA-C₁₆/ODA in HPA aqueous subphase is 0.50 – 0.55 nm^2 . The collapse pressure of OMA-C₁₆/ODA in HPA aqueous solution is 32.6 – 33.4 mN m^{-1} , slightly larger than that in water phase (31.1 mN m^{-1}). Apparently,

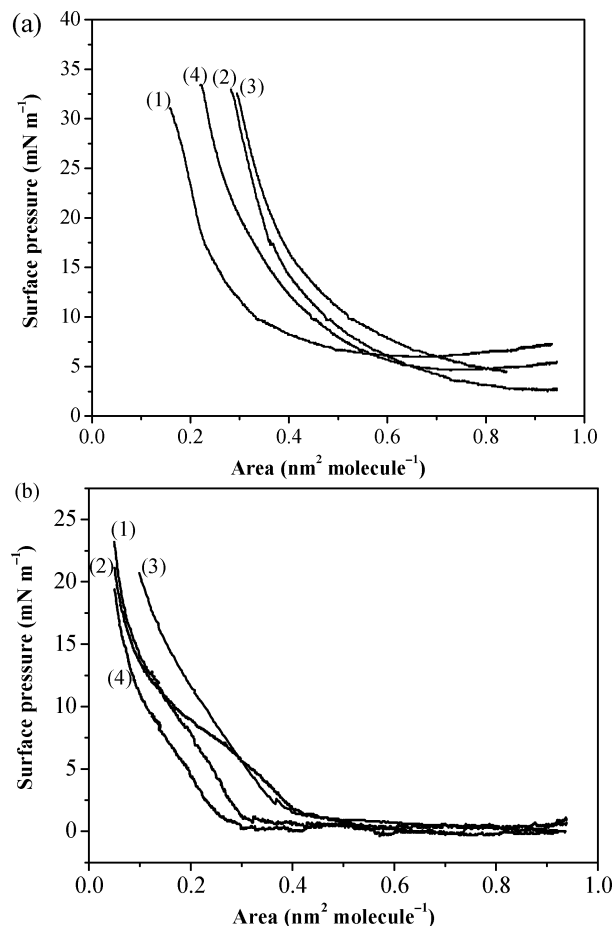


Figure 2. Surface pressure–area (π – A) isotherms of (a) monolayer films of (1) OMA-C₁₆/ODA/H₂O, (2) OMA-C₁₆/ODA/HPMo₁₂, (3) OMA-C₁₆/ODA/HP₂Mo₁₈, and (4) OMA-C₁₆/ODA/HPW₁₂ and (b) monolayer films of (1) OMA-C₈/H₂O, (2) OMA-C₈/HPMo₁₂, (3) OMA-C₈/HP₂Mo₁₈, and (4) OMA-C₈/HAS₂W₁₈.

Table 1. Surface Pressure–Area Isotherm Data of the LB Films

film	solid-state region (mN m^{-1})	cross section ($\text{nm}^2 \text{ molecule}^{-1}$)	collapse pressure (mN m^{-1})
OMA-C ₁₆ /ODA/H ₂ O	17.3–31.1	0.35	31.1
OMA-C ₁₆ /ODA/HPMo ₁₂	17.6–32.9	0.54	32.9
OMA-C ₁₆ /ODA/HP ₂ Mo ₁₈	21.4–32.6	0.55	32.6
OMA-C ₁₆ /ODA/HPW ₁₂	24.9–33.4	0.50	33.4
OMA-C ₈ /H ₂ O	16.8–23.2	0.23	23.2
OMA-C ₈ /HPMo ₁₂	14.6–21.1	0.24	21.1
OMA-C ₈ /HP ₂ Mo ₁₈	15.0–20.7	0.33	20.7
OMA-C ₈ /HAS ₂ W ₁₈	10.5–19.4	0.20	19.4

isotherms on HPA aqueous solutions exhibited an increase in the molecular area and surface pressure compared with the case for pure water, suggesting that HPA molecules are not embedded inside the OMA-C₁₆ molecules (Figure 2a). We note that the curves are steeper at the end of the compression and each of the isotherms is essentially shifted toward larger areas per molecule in the presence of HPA. It was proposed that the HPA molecules are situated in the middle of OMA-C₁₆ and ODA molecules to form alternate lamellar structures with an increased molecular area (vide infra). Compared with the Keggin-type polyoxometalates HPMo₁₂ and HPW₁₂, the Dawson-type species HP₂Mo₁₈ has more charges and bigger volume, which can result in a larger molecular area. Typical isotherms were also observed for the OMA-C₈ case (Figure 2b).

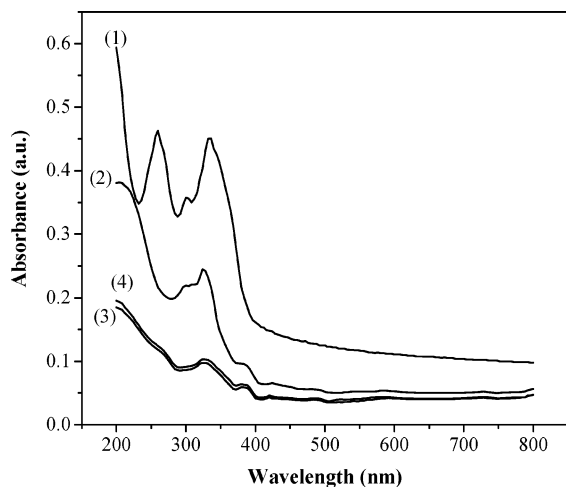


Figure 3. UV-vis spectra of LB films (13 layers) of (1) OMA-C₁₆/ODA/H₂O, (2) OMA-C₁₆/ODA/HPMo₁₂, (3) OMA-C₁₆/ODA/HP₂Mo₁₈, and (4) OMA-C₁₆/ODA/HPW₁₂.

Table 2. Absorption Data for the LB Films

LB film	λ_{\max} (nm)
OMA-C ₁₆ /ODA/H ₂ O	302, 334, 368
OMA-C ₁₆ /ODA/HPMo ₁₂	304, 332, 384
OMA-C ₁₆ /ODA/HP ₂ Mo ₁₈	328, 384
OMA-C ₁₆ /ODA/HPW ₁₂	328, 384
OMA-C ₈ /H ₂ O	312, 344, 360
OMA-C ₈ /HPMo ₁₂	312, 336, 364
OMA-C ₈ /HP ₂ Mo ₁₈	312, 348, 376
OMA-C ₈ /HA ₂ W ₁₈	304, 348, 382

UV-Vis Spectroscopy of LB Films. OMA-C₁₆/ODA/HPA of 13 layers and OMA-C₈/HPA of 9 layers LB films of high morphological stability were successfully deposited onto quartz substrates by the vertical method. The transfer ratio was almost unity in both dipping and lifting processes, indicating the formation of Y-type LB films. As shown in Figure 3, OMA-C₁₆/ODA/H₂O and OMA-C₈/H₂O films display strong absorption bands in the near UV region. These bands are mainly associated with the organic¹($\pi\pi^*$) transitions, possibly with some admixture of mercury atom orbitals, and the 0-0 absorption peak is assigned as the S₀ → S₁ transition.¹⁴ For OMA-C₁₆/ODA/HPA hybrids, the LB films exhibit intense O_d-M (M = Mo, W, 270 nm) and O_{b(c)}-M (330 nm or so) peaks which are perhaps mixed with organic¹($\pi\pi^*$) transitions (where O_d represents the terminal oxygen atoms whereas O_{b(c)} the two different types of bridging oxygen atoms in HPA). Compared with OMA-C₁₆/ODA/H₂O, the absorption peaks of OMA-C₁₆/ODA/HPA LB films are slightly red-shifted which are presumably due to the delocalized π -conjugation caused by OMA and HPA. For OMA-C₈/HPA hybrids, the red shift of the absorption peaks of OMA-C₈/HPA was also observed with respect to that of OMA-C₈/H₂O. See Table 2.

Photoluminescence Properties of LB Films. With this method, luminescent LB films with OMA have been prepared. The photoluminescence (PL) spectra of the LB films (13 layers for OMA-C₁₆ and 9 layers for OMA-C₈) deposited on quartz were measured, with the excitation wavelength at 330 nm (Figure 4). We observe broad intraligand-based¹($\pi\pi^*$) emission peaks in the range of 428-494 nm for each of them which arise from a singlet excited

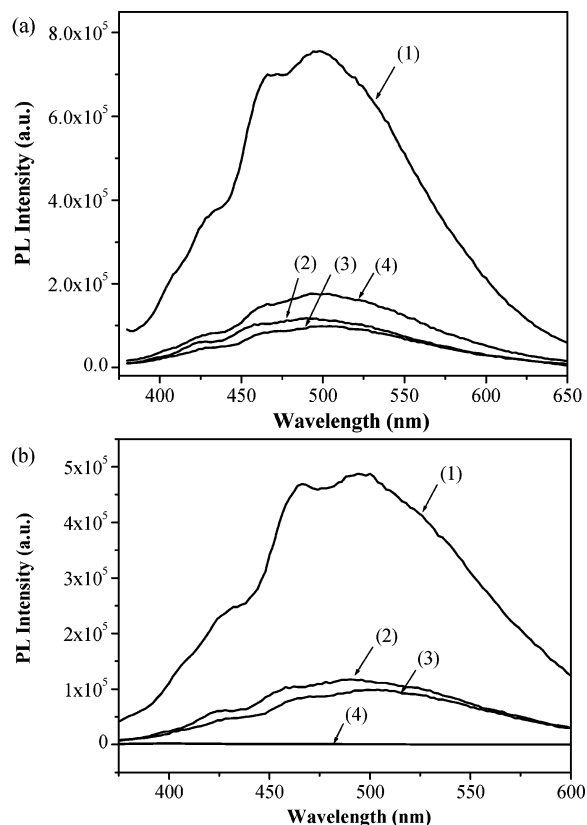


Figure 4. PL spectra of (a) 13-layer LB films of (1) OMA-C₁₆/ODA/H₂O, (2) OMA-C₁₆/ODA/HPMo₁₂, (3) OMA-C₁₆/ODA/HP₂Mo₁₈, and (4) OMA-C₁₆/ODA/HPW₁₂ and (b) 9-layer LB films of (1) OMA-C₈/H₂O, (2) OMA-C₈/HPMo₁₂, (3) OMA-C₈/HP₂Mo₁₈, and (4) OMA-C₈/HA₂W₁₈.

state (S₁). The emission patterns of all the films appear broad and unstructured at room temperature, which are characteristic of an aggregate or excimer site that is caused by the intermolecular mercuriphilic interactions among the neighboring mercury centers.^{14e,15} Such claim was possibly explained by X-ray crystallography and aggregation-induced absorption studies for OMA-C₈. The lattice of OMA-C₈ is characterized by the exhibition of various weak intermolecular Hg...Hg contacts of 3.738-4.183 Å, linking the individual molecular entities together to give a loose polymeric 3D structure (Figure 5). Examination of the absorption behavior of OMA-C₈ in CHCl₃ upon the addition of a nonsolvent MeOH also corroborates the presence of solid-state aggregates in thin films. While the amount of MeOH added is continuously increased at the expense of CHCl₃, a new band in the lower lying regime at 365 nm develops clearly, which corresponds to the lowest energy absorption peak observed in the solid state (Figure 6). Similar solid-state phenomenon is also commonplace for many mercury acetylides,¹⁶ and in several occasions, such noncovalent metallophilicity turns out to be the major driving force toward supramolecular aggregation.¹⁷ Interestingly, inclusion of HPA within the LB films causes important changes in their luminescence properties compared with those observed

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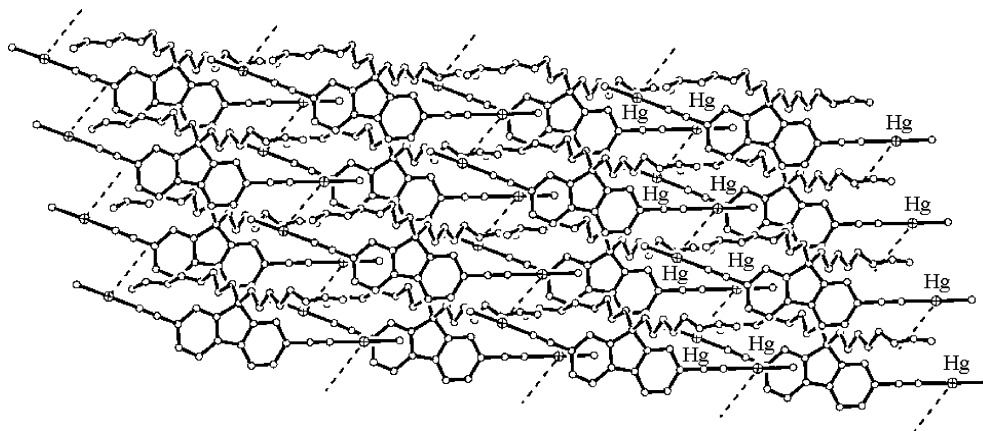


Figure 5. Solid-state packing diagram of OMA-C₈ showing the apparent mercuriphilic interactions among Hg atoms in the crystal lattice.

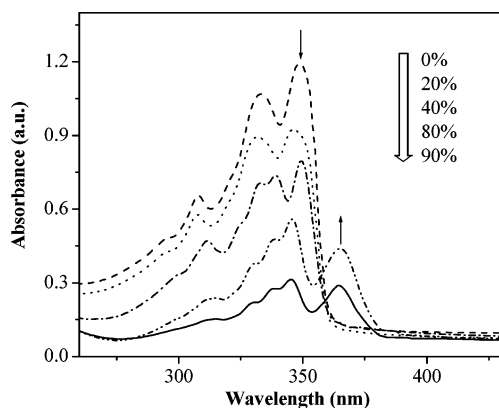


Figure 6. Aggregation-induced absorption spectra of OMA-C₈ at various volume ratios of CHCl₃/MeOH.

on pure water. For hybrid LB films of OMA-C₁₆/ODA/HPA and OMA-C₈/HPA, the PL intensity is lower than that of the OMA-C₁₆/ODA/H₂O and OMA-C₈/H₂O systems, respectively, and it is found that HPA quenches the luminescence of OMA to a certain extent.

Low-Angle X-ray Diffraction. The lamellar structure of the hybrid LB films can be clearly demonstrated by the low-angle X-ray diffraction experiment. Take OMA-C₁₆ as an example; three Bragg peaks are clearly identified in the X-ray diffractogram of a 13-layer OMA-C₁₆/ODA/HPW₁₂ LB film (Figure 7). The peaks at 2.39°, 5.26°, and 8.01° (2θ) correspond to the {001}, {002}, and {003} reflections, respectively, revealing a highly ordered stacking of different layers. According to the Bragg diffraction formula, $n\lambda = 2d \sin \theta$, we can calculate the monolayer average thickness $2/d$ to be 1.85 nm. The periodicity of the layered structure deduced from this experiment is ca. 3.7 nm. It is known that the length of the ODA molecule with a trans zigzag

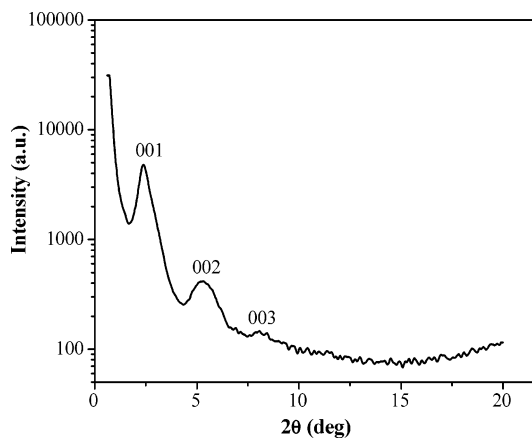


Figure 7. Low-angle X-ray diffraction pattern of the as-prepared OMA-C₁₆/ODA/HPMo₁₂ LB film with 13 layers deposited on glass. Indices for the Bragg reflections are also shown.

conformation is 2.4 nm,¹⁸ whereas those of OMA-C₁₆ and Keggin HPA molecules are 2.2 nm^{14c} and 1 nm,¹⁰ respectively. The total length of the three molecules arranged layer by layer is ca. 5.6 nm, which is larger than the calculated d value (3.7 nm). Our X-ray diffraction results are consistent with the lamellar layered structure for the OMA-C₁₆/ODA/HPMo₁₂ LB film (Figure 8) in which the three molecules lay down at some angles with the interfacial planes.

Atomic Force Microscopy. Surface characterization of the LB films by atomic force microscopy (AFM) has also been performed. The image and structure of the hybrid monolayer film deposited on mica was measured (Figure 9). The AFM image of OMA-C₁₆/ODA/HPMo₁₂ (Figure 9a) shows that the hybrid LB film consists of disperse particles with different domain sizes of 0.60–3.96 nm. The average surface roughness (R_a) is 2.7 nm and the largest roughness (R_z) is 3.6 nm. A dimensionally split phase structure can be observed in which the darker area is mostly filled with HPA and the higher and brighter area is mostly filled with OMA-C₁₆/ODA. The topographic and cross-sectional image of OMA-C₈/HPMo₁₂ monolayer film is shown in Figure 9b with the R_a and R_z values of 0.5 and 14.8 nm, respectively.

Surface Photovoltage Spectroscopy. The photovoltaic effect of the 13-layer hybrid LB films deposited on the silica

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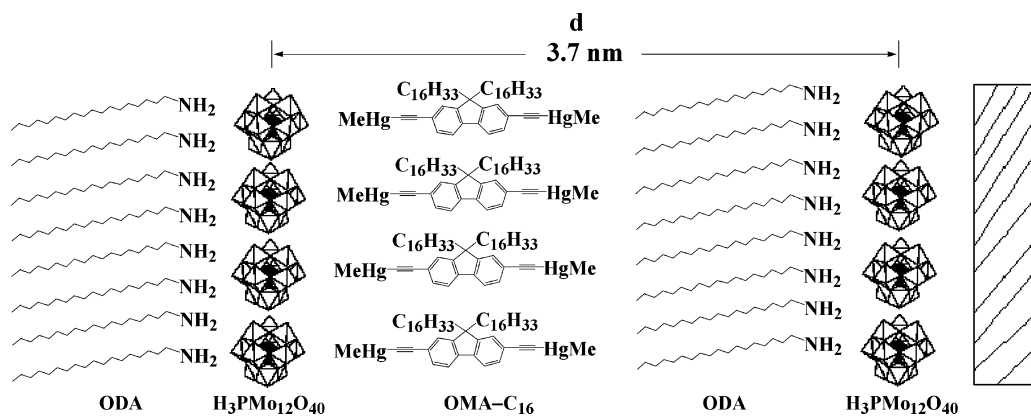


Figure 8. Schematic representation of the proposed structure of the OMA-C₁₆/ODA/H₃PMO₁₂O₄₀ nano hybrid LB film.

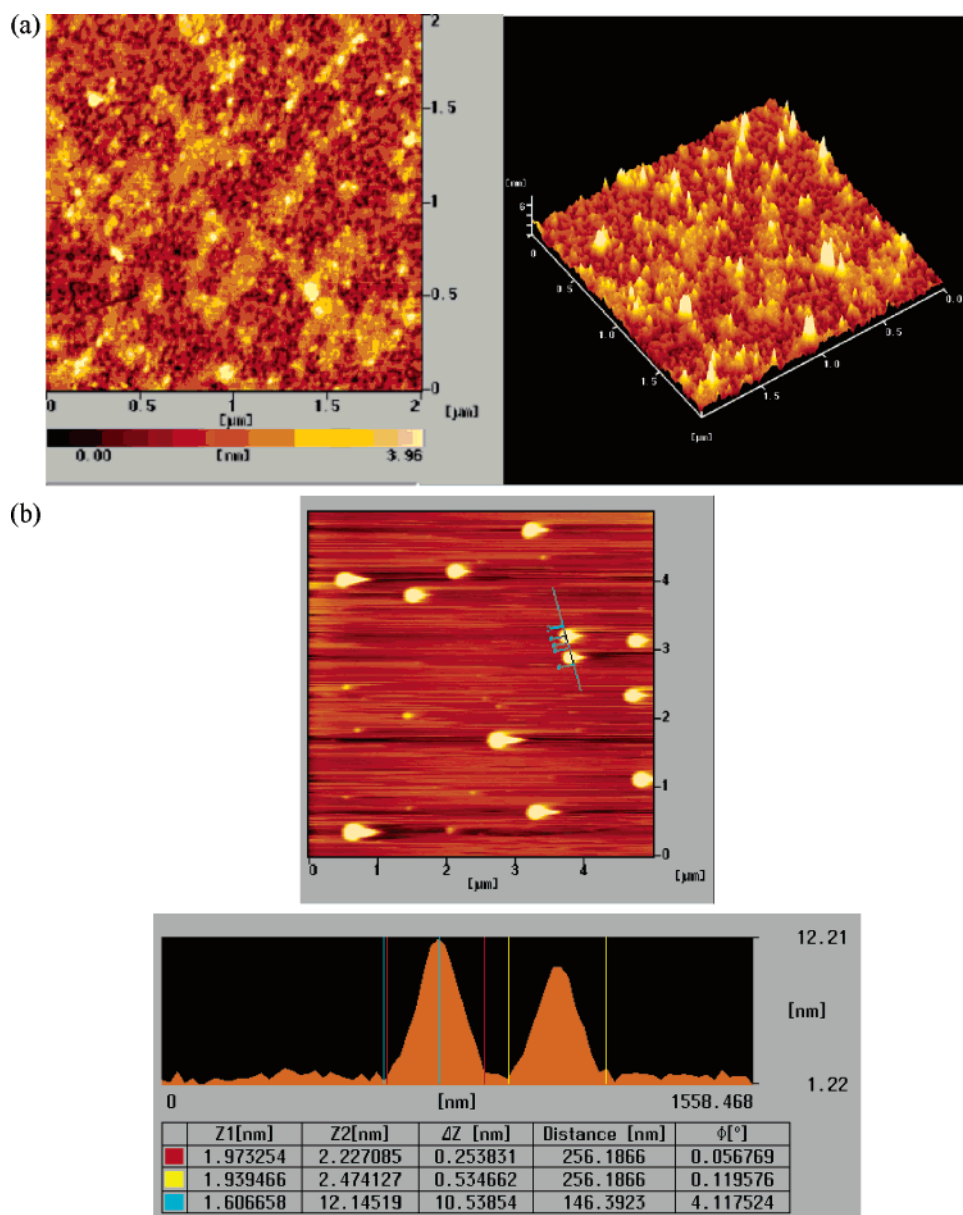


Figure 9. (a) Tapping mode AFM image of the OMA-C₁₆/ODA/HPMO₁₂ monolayer film and (b) AFM topography image (upper) and section profile (lower) of the OMA-C₈/HPMO₁₂ monolayer film.

wafer was also studied after UV-irradiation using the surface photovoltage spectroscopy (SPS) technique. The results are shown in Table 3 and Figure 10. The impact of mercury acetylides in photovoltaic phenomenon is well-illustrated in

the present study. The OMA-C₁₆/ODA/H₂O and OMA-C₁₆/ODA/HPA hybrid LB films display strong photovoltage effect in the near UV region. The photovoltage performance generally follows the order OMA-C₁₆/ODA/HPMO₁₂ >

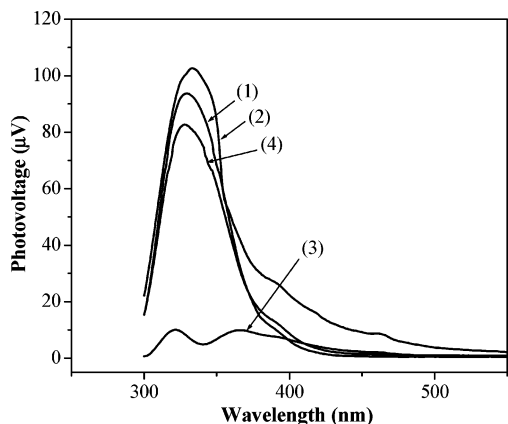


Figure 10. Surface photovoltage responses of hybrid LB films (13 layers) of (1) OMA-C₁₆/ODA/H₂O, (2) OMA-C₁₆/ODA/HPMo₁₂, (3) OMA-C₁₆/ODA/HP₂Mo₁₈, and (4) OMA-C₁₆/ODA/HPW₁₂.

Table 3. Surface Photovoltage Spectroscopic Data for Various LB Films

LB film	λ (nm)	V (μ V)
OMA-C ₁₆ /ODA/H ₂ O	329	93.7
OMA-C ₁₆ /ODA/HPMo ₁₂	333	102.7
OMA-C ₁₆ /ODA/HP ₂ Mo ₁₈	321	10.1
	365	9.9
OMA-C ₁₆ /ODA/HPW ₁₂	328	82.7

OMA-C₁₆/ODA/H₂O > OMA-C₁₆/ODA/HPW₁₂ > OMA-C₁₆/ODA/HP₂Mo₁₈ and the photovoltage responses can be as high as 102.7 μ V. It is known that mercury acetylide complex which contains π -conjugated electrons can serve as a σ - π electron donor, whereas the three types of HPA are photosensitive substances and act as electron acceptors. With this concept, our OMA/ODA/HPA hybrid system can manifest donor- π -acceptor (D- π -A) structural motif and the π -conjugated system can be used as an electron-transporting passage. When the LB film was excited by light, the photovoltaic effect readily comes into play. The structure of the film does permit electron delocalization to an appreciable extent. It was found that a change in the metal type from Mo to W ((2) vs (4) in Figure 10) caused a noticeable decrease in the photovoltage at a given wavelength. Moreover, heteropolyoxomolybdates of the Keggin-type exhibit the highest photovoltage values while the Dawson-type gives the lowest photovoltage response ((2) vs (3) in Figure 10). Presumably, the rugby-shaped HP₂Mo₁₈ tends to be less effective in self-organizing with OMA relative to the more spherical-like HPMo₁₂. In addition, it is believed that charge transfer can take place between the interface of the title LB film and *p*-Si substrate.¹⁹ The adsorption of polyoxometalate anions along a ODA monolayer must be induced by electrostatic interactions between the ion and the charge density at the interface. Accordingly, free-standing films containing alkynylmercury complexes can be prepared by electrostatic layer-by-layer assembly.

Electrical Characterization of Monolayer LB Films by Scanning Tunneling Microscopy. The electric conductivity behavior of OMA-C₈/HP₂Mo₁₈ monolayer film on silica

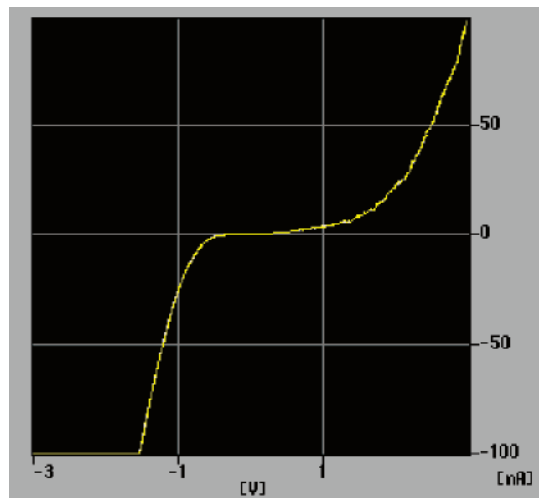


Figure 11. I - V curve of OMA-C₈/HAs₂W₁₈ monolayer film on the silica wafer.

wafer was examined using scanning tunneling microscopy. When the voltage is monitored at ± 10 V, the tunneling current obtained is 0.08 to ca. -0.92 nA. This value only shows a slight increase as compared with that for OMA-C₈/H₂O (0 to ca. -0.92 nA). Remarkably, for OMA-C₈/HAs₂W₁₈ monolayer film, it can show good electrical conductivity, and the tunneling current amounts to ± 100 nA when the voltage is set at ± 3 V. A representative I - V plot is shown in Figure 11 for such a device. It should be noted that OMAs are very poor conductors in the intrinsic state but our hybrid LB composite in the device can be semiconducting. The largest conductance of the device is about 0.17×10^{-6} S at -1.6 V.

Concluding Remarks

This study has demonstrated that mercury acetylides can be used to build new organometallic/inorganic superlattices useful for photoelectric applications. These compounds can form steady monolayers and multilayers at the interface of air and heteropolyacid aqueous solutions. This is the first observation of periodicity of OMA molecules in built-up LB films. Because of the charge transfer between mercury acetylide donor and heteropolyacid acceptor as well as the possible hydrogen-bonding network formation and adsorption phenomenon, the films can be significantly compressed and aggregated in the solid region and Y-type photovoltaic nano-hybrid LB films have been prepared. By virtue of the D- π -A structure for the films, photoconductivity was easily observed upon photoexcitation. Organomercury acetylides allow a delocalized electron transfer in the large π -conjugated bond system, leading to good electrical conductivity in the LB films.

For the hybrid LB films based on OMA, weak d¹⁰...d¹⁰ mercuriphilic interaction, π - π^* transition between the triple bond of OMA and the aromatic system of the ligand, and electron transfer between electron donor and acceptor are believed to be the main causes for the good photovoltaic effect of the title hybrid LB films. Such an organic/organometallic/inorganic multifunctional assembly gives rise to new ultrathin materials. The work can be

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further extended to different kinds of polyoxometalate clusters, possibly affording new lamellar materials. In fact, a correct choice of polyoxometalate allows the elaboration of LB films exhibiting particular surface and optoelectronic properties. Precise construction of a thin film of polyoxometalate is desired for realizing practical molecular functional devices. Obviously, the occurrence of photoluminescent activity confirms the potential for

creating light-emitting multilayer films with OMA and HPA.

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